

LOW BORON E-GLASS COMPOSITION

BACKGROUND OF THE INVENTION

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Field of the Invention

The present invention relates to a low boron containing E-glass composition. More specifically, the present invention relates to a glass composition which comprises a small amount of boron, and still maintains fiberization capability.

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Description of the Related Art

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For many years, the industry has prepared fiberglass and fiberglass products with a base glass composition of the "E-glass" type. "E" glass is generally described in U.S. Patent No. 2,344,961, and the described glasses are typified by the composition listed below (with the constituents being in weight percent):

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SiO_2	52-56%
Al_2O_3	12-16%
CaO	16-19%
MgO	3-6%
B_2O_3	9-11%

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U.S. Patent No. 2,571,074 describes a "modified E" glass, which is typically devoid of MgO, and with the CaO content much higher than a typical "E" glass. In general, the glasses have a composition as shown below (the percentages being weight percent):

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SiO_2	52-56%
Al_2O_3	12-16%
CaO	19-25%
B_2O_3	8-13%

U.S. Patent No. 4,166,747 describes an E glass which attempts to lower the B_2O_3 content in the glass used to manufacture glass fibers in order to achieve reductions in batch costs. To compensate, the glass composition used has only a

small quantity of MgO. In general, the glass has a composition as shown below (the percentages being in weight percent):

5	SiO ₂	54-55%, more preferably 54.2-55%
	Al ₂ O ₃	13-14%
	CaO	22-24%, more preferably 22.1-23.2%
	MgO	0.2-0.5%, more preferably 0.3-0.4%
	B ₂ O ₃	5.5-6.8%, more preferably 6.5-6.8%

10 Thus, typical E-glass compositions used in the industry generally contained approximately 6% or more B₂O₃. As a result of this level of B₂O₃, abatement of E-glass furnaces is required to reduce particulate emissions into the environment. As a result of the challenging economics, and environmental issues, E-glass type fibers have been proposed that contain little or no B₂O₃, i.e., see, for example, WO
15 00/73231; WO 01/32576; U.S. Patent No. 6,136,735 and U.S. Patent No. 5,789,329. Benefits are claimed associated with lower abatement costs and lower batch costs due to boron-free base glass formulations.

 The elimination of B₂O₃ in the base glass compositions, however, has resulted in increases in the high temperature viscosity (HTV) and liquidus
20 temperature relative to standard E-glass containing 6% or more B₂O₃. HTV is the temperature at which the glass melt has a viscosity of 1000 poises. In the production of fiberglass, the glass melt in the bushing is typically maintained at 50-100°F above the HTV for optimum fiberization efficiency. A glass composition with a higher HTV requires more energy to achieve this viscosity and therefore, incurs higher energy
25 costs. The liquidus temperature is the highest temperature at which devitrification (crystallization) of the glass melt occurs. In fiberglass production, it is greatly preferred that the glass melt be maintained at a temperature at least 150°F above the liquidus to avoid crystallization problems and consequently, lower fiberization efficiencies. It has been found that in the commercial production of E-glass fibers,
30 fiberization efficiency is significantly reduced as the difference between HTV and liquidus (referred to as "delta") falls below approximately 150°F.

 Thus, it would be desirable to take advantage of the cost benefits associated with lower boron E-glass formulations, but avoid the disadvantages of crystallization and fiberization problems. The challenge in producing such a composition is
35 developing the right combination of oxides that yields a glass with a suitable HTV and

maintains a delta of at least 150°F, while utilizing a lower amount of boron. Providing such a glass composition would be of exceptional value to the industry.

Accordingly, it is an object of the present invention to provide such a glass composition which can be used to make useful continuous fiberglass strands, which process is more economic to make, yet avoids major manufacturing problems associated with crystallization of the glass.

This and other objects of the present invention will become apparent to the skilled artisan upon a review of the following description and the claims appended thereto.

SUMMARY OF THE INVENTION

In accordance with the foregoing objective, provided by the present invention is a glass composition having a reduced amount of boron which is useful in preparing continuous fiberglass strands. The composition allows one to use a minimal amount of boron, yet provides a glass with a suitable HTV and a delta of at least 150°F above the liquidus temperature to avoid crystallization problems. The composition is thus cost effective and practical for use in preparing E glass continuous strands.

The glass composition has an HTV of no greater than 2300°F and a liquidus temperature of at least 150°F less than the HTV temperature. The composition comprises the following ingredients:

SiO ₂	52-62 wt. %
B ₂ O ₃	3.5-5.5 wt. %
CaO	18-25 wt. %
MgO	0.5-4 wt. %
Al ₂ O ₃	10-15 wt. %
Na ₂ O	0.25-2 wt. %;

It has been found that glass compositions employing the foregoing percentages can achieve a suitable HTV of less than 2300°F so that the process remains energy efficient and extraordinary bushings capable of withstanding higher fiberization temperatures are not needed. Furthermore, the glass compositions within the foregoing percentages permit a difference between the HTV and liquidus temperature of at least 150°F, thereby permitting fiberization efficiency. While enjoying the foregoing benefits, the compositions also reduce batch material costs

considerably, as well as the abatement requirements due to the lower amount of B_2O_3 contained in the glass composition.

DETAILED DESCRIPTION OF THE PREFERRED EMBODIMENTS

5 The glass compositions of the present invention are characterized by an amount of B_2O_3 which is less than that generally employed in industrial E glass compositions, yet more than that contained in the compositions that have evolved containing substantially no boron. Thus, the glass compositions of the present invention allows one to avoid the problems of fiberization efficiency and the need to
10 use substantially higher fiberizing temperatures, which can do damage to the bushing employed. To achieve good, continuous fiberglass strands employing this lower amount of B_2O_3 , and to have a suitable HTV of less than 2300°F and a liquidus temperature at least 150°F less than the HTV temperature, it has been found that the glass composition must also employ SiO_2 , CaO , MgO , Al_2O_3 and Na_2O within the
15 following weight percentage ranges:

	SiO_2	52-62 wt. %
	B_2O_3	3.5-5.5 wt. %
	CaO	18-25 wt. %
20	MgO	0.5-4 wt. %
	Al_2O_3	10-15 wt. %
	Na_2O	0.25-2 wt. %;

25 The reduced B_2O_3 content permits one to realize cost savings associated with lower batch costs and abatement requirements. At the same time, employing the foregoing glass composition allows one to also achieve the cost savings without significantly increasing the energy required for melting the glass, a reduction in fiberization efficiency, or requiring a development of new bushing technology.

30 While the foregoing compositions must contain the ingredients noted, other ingredients may be present in the composition of ingredients. Such impurities are generally TiO_2 and Fe_2O_3 . In general, glass will contain TiO_2 in a range of from 0.4 to 0.6 wt. % as an impurity, and Fe_2O_3 in a range of from 0.2 to 0.3 wt. % as an impurity. As well, a small amount of K_2O can be present as an impurity, for example, up to 0.5 wt. %. Generally, no Li_2O or F_2 is found in the glasses of the present
35 invention. Thus, while the glass composition described herein will relate to those

ingredients important to achieving the objectives of the present invention, one of ordinary skill in the art will understand that impurities generally found in glass such as TiO_2 and Fe_2O_3 will be present in the glass to a limited extent.

In a preferred embodiment, the glass compositions of the present invention
5 more preferably contain SiO_2 in an amount ranging from about 53 to 57 weight percent; an amount of B_2O_3 more preferably in the range of from about 4 to 5 weight percent; an amount of CaO more preferably in the range of from about 20-24 weight percent; an amount of MgO more preferably in the range of from about 1-3.5 weight percent; an amount of Al_2O_3 in the range of from about 12-15 weight percent; and an
10 amount of Na_2O in the range of from about 0.5-1.5 weight percent. Thus, a preferred composition would contain the following amounts of ingredients;

	SiO_2	53-57 wt. %
	B_2O_3	3-5 wt. %
15	CaO	20-24 wt. %
	MgO	1-3.5 wt. %
	Al_2O_3	12-15 wt. %
	Na_2O	0.5-1.5 wt. %.

20 Most preferably, the amount of SiO_2 ranges from about 54-56 weight percent; the amount of B_2O_3 most preferably ranges from about 4.5 to 5.0 weight percent; the amount of CaO ranges from about 21-23.5 weight percent; the most preferred amount of MgO ranges from about 1.5-3 weight percent; and the most preferred amount of Al_2O_3 ranges from about 14-15 weight percent; and the amount of Na_2O
25 ranges from about 1.0-1.5 weight percent.

In selecting the particular ingredients for the glass composition within the foregoing ranges, the HTV realized is generally less than 2300°F. Easily, however, the glass composition can be formulated to achieve an HTV less than 2250°F, and most preferably less than 2200°F. The liquidus temperature of the glass composition
30 is at least 150°F less than that of the HTV temperature, to therefore provide a delta T of 150°F. Thus, the liquidus temperature is generally at least below 2150°F, more preferably less than 2100°F, and most preferably less than 2000°F. Also, it is preferred that the delta T is also greater than 160°F.

The reduction of B_2O_3 cannot simply be compensated for by the increase of
35 another oxide in the glass composition. The combination of changes in the oxides is

complex, and must be within the ranges discussed above. This selection must be made with an eye toward the HTV and liquidus temperature of the composition, for otherwise the advantages of the present invention are lost. Generally, selecting and mixing oxides within the ranges discussed above will provide one with a glass

- 5 composition that meets the HTV, liquidus and delta requirements for economically and efficiently preparing continuous fiberglass strands.

In order to further illustrate the present invention, several experimental runs were conducted in which glass compositions were experimentally melted and tested for HTV, liquidus and chemical composition. As well, cost savings per ton as

- 10 opposed to typical commercial E glass was calculated. Table 1 below sets forth the specific examples of glass compositions.

TABLE 1
Examples of glass composition and properties
for lower B₂O₃-containing E-glass fibers

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<u>Oxide (weight %)</u>	<u>A</u>	<u>B</u>	<u>C</u>	<u>D</u>
SiO ₂	55.82	56.30	56.70	57.45
Al ₂ O ₃	13.21	13.20	13.22	13.21
CaO	21.56	21.38	21.87	21.03
B ₂ O ₃	5.24	4.70	4.24	3.65
Na ₂ O	1.30	1.31	1.31	1.31
MgO	2.05	2.34	1.86	2.53
TiO ₂	0.54	0.54	0.53	0.54
Fe ₂ O ₃	0.22	0.22	0.22	0.23
K ₂ O	0.03	0.03	0.03	0.03
HTV (°F)	2172	2179	2195	2217
Liquidus (°F)	1956	1988	1999	2048
Delta (°F)	216	191	196	169
Cost savings/ton vs. E-glass	\$6.0	\$9.8	\$14.2	\$18.0

- 20 It is clear from these experiments that in order to reduce the B₂O₃ concentration in the glass formulation, it cannot only be substituted for by a single

oxide (e.g., 3% B₂O₃ vs. 3% SiO₂). In order to maintain the required HTV and liquidus temperature, a combination of oxides must replace B₂O₃ in the composition. Also, as the HTV increases over that of standard E-glass which is common for those compositions of the prior art which contain substantially no B₂O₃, and the same bushings are used for fiberization, bushing lifetime decreases and therefore adds costs that are not factored into the cost savings listed in Table 1, which savings are realized with the compositions of the present invention.

Other examples of suitable compositions within the present invention which provide the advantages of a suitable HTV and a delta T are shown below in Table 2.

Table 2

Example	SiO ₂	Al ₂ O ₃	B ₂ O ₃	Na ₂ O	CaO	MgO	HTV°F	Liq.°F	Delta T
1	53.35	14.27	5.50	1.43	22.82	1.78	2133.4	1983.4	150
2	60.61	11.87	4.5	0.25	18.63	3.29	2288.0	2130.6	157.4
3	58.45	15.00	5.00	1.74	18.32	0.64	2289.9	1936.1	353.8
4	58.07	15.00	5.00	1.59	18.70	0.80	2282.5	1923.7	358.8
5	60.29	11.61	5.50	0.25	18.40	3.10	2300.0	2150.0	150
6	60.72	11.96	4.25	0.25	18.73	3.24	2290.4	2120.1	170.3
7	58.00	13.66	4.75	2.00	18.17	2.57	2259.6	2054.2	205.4
8	54.34	15.00	4.00	0.25	24.99	0.57	2256.0	2106.0	150.0
9	58.00	13.66	4.75	2.0	18.17	2.57	2259.6	2054.2	205.4
10	57.39	14.92	3.75	2.0	18.64	2.45	2298.2	2117.3	180.9
11	58.87	14.55	3.50	1.60	20.13	0.50	2293.7	2029.0	264.7
12	57.97	13.77	5.25	0.5	18.00	3.66	2232.7	2082.7	150.0

While the present invention has been described in accordance with certain preferred embodiments, it will be apparent to one of ordinary skill in the art that many changes and modification can be made thereto without departing from the spirit or scope of the invention as set forth herein.